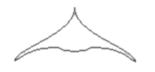
Introduction to Trace

Dan Retief PhD



Aero Design Limited



Composites Design Challenge

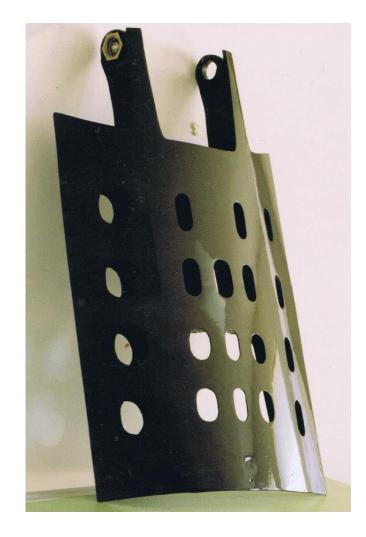
Simplify the way we deal with material properties and make it as easy as working with Aluminium.

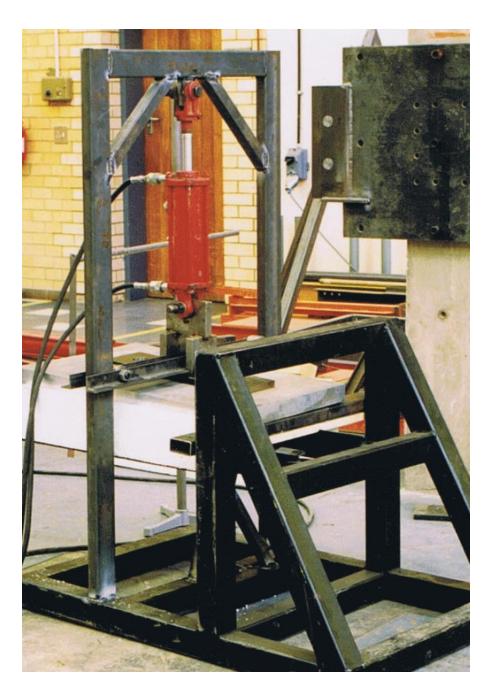
Lets see where we were before:



Black Aluminium

Design it as if it is aluminium or replace existing aluminium component: Extensive Use of Quasi Isentropic Used rules of thumb to design equivalent laminates Use large factors of safety

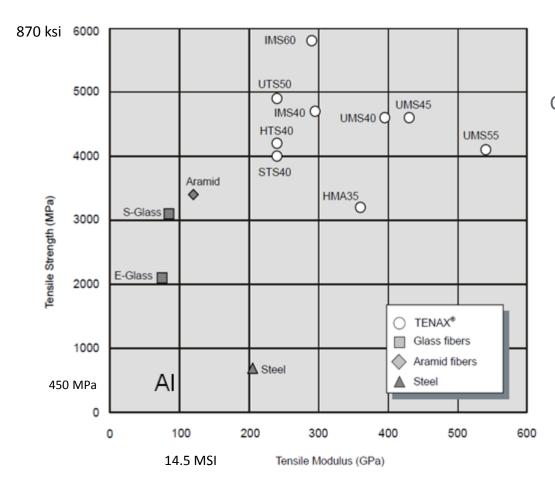




Test Outcomes

Strength and stiffness determined by test. Weight similar to aluminium. Sometimes too heavy. Stiffness barely better than Aluminium. Cost too high to redesign and retest. What is the point?

PRODUCTS LINEUP OF TENAX® FILAMENT



The Potential

Carbon Filaments with stiffness of steel Strength over 10X that of aluminium Stiffness over 2X that of aluminium Corrosion free Fatigue Resistant Efficiency of Sandwich Structures

> Some challenges: Temperature Moisture Damage Tolerance & resistance

Prototype Cargo Pod



Ultimate Static Test > 6000 lb



Some success

- Sandwich Construction
- Weight of pod 45kg
- Loaded with over 6000 lb of lead ingots
- No failure, high stiffness
- High strength to weight ratio
- Achieved with glass fibre facings on Nomex Honeycomb core

The potential is there...





Hopper Gate Box

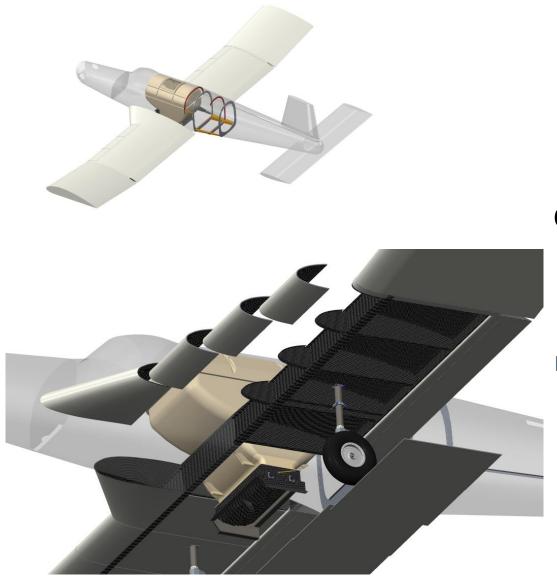
Carbon – 15kg







Steel – 65kg



Cresco Carbon Composite Centre Wing Section

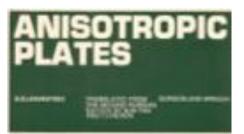
New Project, New Design Challenges

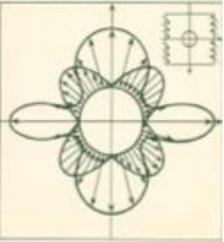




Trace to the rescue!

Beyond Black Aluminium







Online, Live Composites Design Workshop X Stephen W. Tsai & Associates

July 13-17, 2015; noon to 4 PM PST - USA



Stephen W. Tsai José Daniel D. Melo

"A must-have book with logic and ingenuity"

- Truly authoritative and practical.
- 20 hours of live online sessions.

• Presents a totally new invariantbased approach to stiffness and strength and the concept of homogenization for optimal design. Much simplified.

• Sessions including tools and case studies are downloadable for later individual viewing.

• US\$1,500 registration* covers: the new book *Composite Materials Design and Testing*, two e-books, tools, composites app iMicMac and iPad Air 2.

• Optional official transcript of records and certificate from Stanford University**.

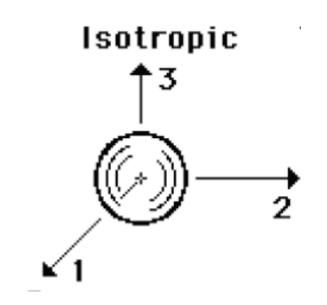
For information on ordering book and workshop registration: http://compositesdesign.stanford.edu

* The registration fee is \$1000* for participants living outside the USA and does not include iPad. ** Continuing Studies courses carry Continuing Education Units (CEU's), not undergraduate/graduate credits. Credits cannot be applied toward any Stanford degree. Credits will be recorded on your transcript and you might be able to transfer the credits to another university, subject to that university's policies.

LAMINATED PLATE THEORY: Stiffness

Composites Design Workshop X

Stephen W. Tsai Stanford University July 13, 2015



Hooke's Law

In matrix form, Hooke's law for isotropic materials can be written as:

$$\begin{bmatrix} \varepsilon_{11} \\ \varepsilon_{22} \\ \varepsilon_{33} \\ 2\varepsilon_{23} \\ 2\varepsilon_{13} \\ 2\varepsilon_{13} \\ 2\varepsilon_{12} \end{bmatrix} = \begin{bmatrix} \varepsilon_{11} \\ \varepsilon_{22} \\ \varepsilon_{33} \\ \gamma_{23} \\ \gamma_{13} \\ \gamma_{12} \end{bmatrix} = \begin{bmatrix} 1 & -\nu & -\nu & 0 & 0 & 0 \\ -\nu & 1 & -\nu & 0 & 0 & 0 \\ -\nu & -\nu & 1 & 0 & 0 & 0 \\ 0 & 0 & 0 & 2(1+\nu) & 0 & 0 \\ 0 & 0 & 0 & 0 & 2(1+\nu) & 0 \\ 0 & 0 & 0 & 0 & 0 & 2(1+\nu) \end{bmatrix} \begin{bmatrix} \sigma_{11} \\ \sigma_{22} \\ \sigma_{33} \\ \sigma_{23} \\ \sigma_{13} \\ \sigma_{12} \end{bmatrix}$$

where $\gamma_{ij}:=2arepsilon_{ij}$ is the **engineering shear strain**. The inverse relation may be written as

$$\begin{bmatrix} \sigma_{11} \\ \sigma_{22} \\ \sigma_{33} \\ \sigma_{23} \\ \sigma_{13} \\ \sigma_{12} \end{bmatrix} = \frac{E}{(1+\nu)(1-2\nu)} \begin{bmatrix} 1-\nu & \nu & \nu & 0 & 0 & 0 \\ \nu & 1-\nu & \nu & 0 & 0 & 0 \\ \nu & \nu & 1-\nu & 0 & 0 & 0 \\ 0 & 0 & 0 & (1-2\nu)/2 & 0 & 0 \\ 0 & 0 & 0 & 0 & (1-2\nu)/2 & 0 \\ 0 & 0 & 0 & 0 & 0 & (1-2\nu)/2 \end{bmatrix} \begin{bmatrix} \varepsilon_{11} \\ \varepsilon_{22} \\ \varepsilon_{33} \\ \varepsilon_{23} \\ \varepsilon_{23} \\ \varepsilon_{23} \\ \varepsilon_{21} \\ \varepsilon_{23} \\ \varepsilon_{24} \\ \varepsilon_{25} \\ \varepsilon_$$

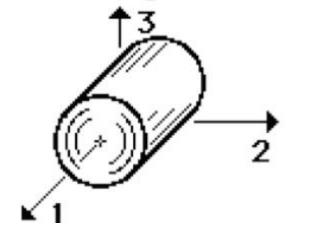
Only two independent variables: E, v

Under plane stress $\sigma_{31} = \sigma_{13} = \sigma_{32} = \sigma_{23} = \sigma_{33} = 0$ $\begin{bmatrix} \varepsilon_{11} \\ \varepsilon_{22} \\ 2\varepsilon_{12} \end{bmatrix} = \frac{1}{E} \begin{bmatrix} 1 & -\nu & 0 \\ -\nu & 1 & 0 \\ 0 & 0 & 2(1+\nu) \end{bmatrix} \begin{bmatrix} \sigma_{11} \\ \sigma_{22} \\ \sigma_{12} \end{bmatrix}$

Or in inverse form the familiar:

$$\begin{bmatrix} \sigma_{11} \\ \sigma_{22} \\ \sigma_{12} \end{bmatrix} = \frac{E}{1 - \nu^2} \begin{bmatrix} 1 & \nu & 0 \\ \nu & 1 & 0 \\ 0 & 0 & \frac{1 - \nu}{2} \end{bmatrix} \begin{bmatrix} \varepsilon_{11} \\ \varepsilon_{22} \\ 2\varepsilon_{12} \end{bmatrix}$$

Transverly Isotropic



The material matrix $\underline{\underline{K}}$ has a symmetry with respect to a given orthogonal transformation (\underline{A}) if it does not change when subjected to that transformation. For invariance of the material properties under such a transformation we require

$$A \cdot \mathbf{f} = K \cdot (A \cdot d) \implies \mathbf{f} = (A^{-1} \cdot K \cdot A) \cdot d$$

Hence the condition for material symmetry is (using the definition of an orthogonal transformation)

$$\boldsymbol{K} = \boldsymbol{A}^{-1} \cdot \boldsymbol{K} \cdot \boldsymbol{A} = \boldsymbol{A}^T \cdot \boldsymbol{K} \cdot \boldsymbol{A}$$

Orthogonal transformations can be represented in Cartesian coordinates by a 3 imes3 matrix $oldsymbol{A}$ given by

$$\underline{\underline{A}} = \begin{bmatrix} A_{11} & A_{12} & A_{13} \\ A_{21} & A_{22} & A_{23} \\ A_{31} & A_{32} & A_{33} \end{bmatrix}$$

Therefore the symmetry condition can be written in matrix form as

$$\underline{\underline{K}} = \underline{\underline{A}}^T \underline{\underline{K}} \underline{\underline{A}}$$

For a transversely isotropic material, the matrix \underline{A} has the form

$$\underline{\underline{A}} = \begin{bmatrix} \cos\theta & \sin\theta & 0\\ -\sin\theta & \cos\theta & 0\\ 0 & 0 & 1 \end{bmatrix}$$

In linear elasticity, the stress and strain are related by Hooke's law:

$\underline{\underline{\sigma}} = \underline{\underline{\mathsf{C}}} \, \underline{\underline{\varepsilon}}$

or, using Voigt notation

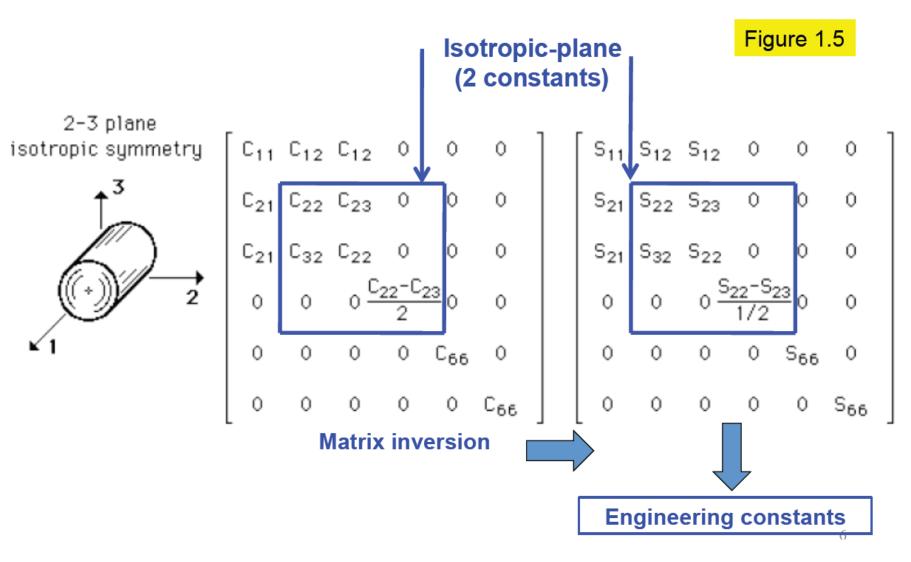
σ_1	C_{11}	C_{12}	C_{13}	C_{14}	C_{15}	C_{16}	ε_1
σ_2	C_{12}	C_{22}	C_{23}	C_{24}	C_{25}	C_{26}	ε_2
σ_3	C_{13}	C_{23}	C_{33}	C_{34}	C_{35}	C_{36}	ε_3
σ_4	C_{14}	C_{24}	C_{34}	C_{44}	C_{45}	C_{46}	ε_4
σ_5	C_{15}	C_{25}	C_{35}	C_{45}	C_{55}	C_{56}	ε_5
$\begin{bmatrix} \sigma_1 \\ \sigma_2 \\ \sigma_3 \\ \sigma_4 \\ \sigma_5 \\ \sigma_6 \end{bmatrix} =$	C_{16}	C_{26}	C_{36}	C_{46}	C_{56}	C_{66}	ε_6

The condition for material symmetry in linear elastic materials is

 $\underline{\underline{C}} = \underline{\underline{A}_{\varepsilon}}^{T} \underline{\underline{C}} \underline{\underline{A}_{\varepsilon}} \qquad \text{giving:}$

$$\underline{\boldsymbol{\varsigma}} = \begin{bmatrix} C_{11} & C_{12} & C_{13} & 0 & 0 & 0 \\ C_{12} & C_{11} & C_{13} & 0 & 0 & 0 \\ C_{13} & C_{13} & C_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & (C_{11} - C_{12})/2 \end{bmatrix} = \begin{bmatrix} C_{11} & C_{11} - 2C_{66} & C_{13} & 0 & 0 & 0 \\ C_{11} - 2C_{66} & C_{11} & C_{13} & 0 & 0 & 0 \\ C_{13} & C_{13} & C_{33} & 0 & 0 & 0 \\ 0 & 0 & 0 & C_{44} & 0 & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & C_{44} & 0 \\ 0 & 0 & 0 & 0 & 0 & C_{66} \end{bmatrix}$$

Stiffness and Compliance Matrices

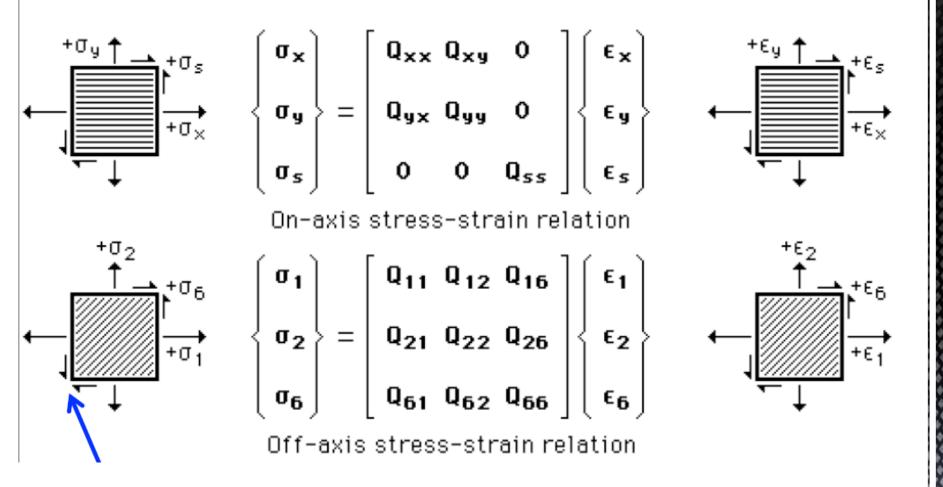


In engineering notation,

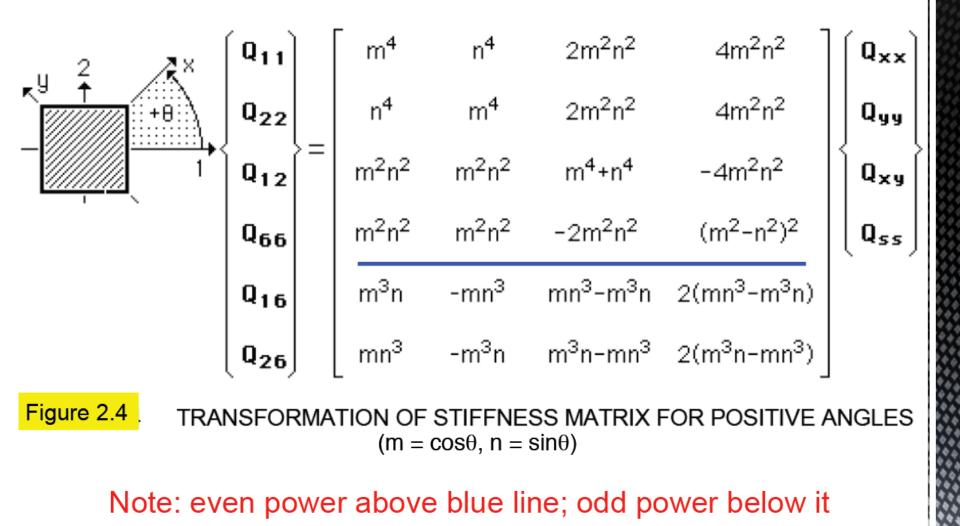
$$\underline{\underline{C}}^{-1} = \begin{bmatrix} \frac{1}{E_{x}} & -\frac{\nu_{yx}}{E_{x}} & -\frac{\nu_{zx}}{E_{x}} & 0 & 0 & 0\\ -\frac{\nu_{xy}}{E_{x}} & \frac{1}{E_{x}} & -\frac{\nu_{zx}}{E_{z}} & 0 & 0 & 0\\ -\frac{\nu_{xx}}{E_{x}} & -\frac{\nu_{xx}}{E_{x}} & \frac{1}{E_{z}} & 0 & 0 & 0\\ 0 & 0 & 0 & \frac{1}{G_{yz}} & 0 & 0\\ 0 & 0 & 0 & 0 & \frac{1}{G_{yz}} & 0\\ 0 & 0 & 0 & 0 & 0 & \frac{2(1+\nu_{xy})}{E_{x}} \end{bmatrix}$$

Note we have 5 variables, compared with 2 in isotropic material.

In the plane stress case for stiffness |Q|



Plane Stress Stiffness Transformation



Linear Combinations of [Q]

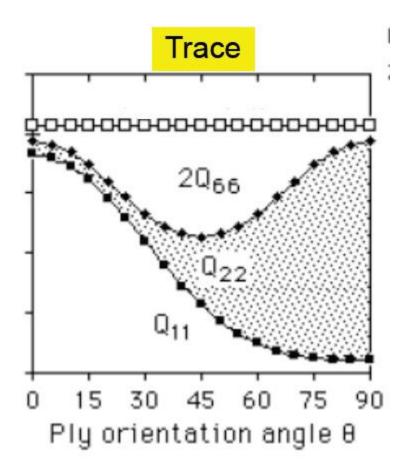
 $m^{4} = \frac{3 + \cos 2\theta + \cos 4\theta}{8}, \quad m^{3}n = \frac{2\sin 2\theta + \sin 4\theta}{8}$ $m^{2}n^{2} = \frac{1 - \cos 4\theta}{8}, \quad mn^{3} = \frac{2\sin 2\theta - \sin 4\theta}{8}, \quad n^{4} = \frac{3 - 4\cos 2\theta + \cos 4\theta}{8}$ (3.7)

 Table 2.1
 LINEAR COMBINATIONS OF ON-AXIS STIFFNESS MODULI

	Q _{××}	Qyy	Q _{×y}	Q _{ss}	Invariant?
$U_1 = U_4 + 2U_5$	3/8	3/8	1/4	1/2	Yes
U ₂	1/2	-1/2	0	0	No
U3	1/8	1/8	-1/4	-1/2	No
$U_4 = U_1 - 2U_5$	1/8	1/8	3/4	-1/2	Yes
$U_5 = (U_1 - U_4)/2$	1/8	1/8	-1/4	1/2	Yes

Two independent invariants: $U_1 = U_4 + 2U_5$

20

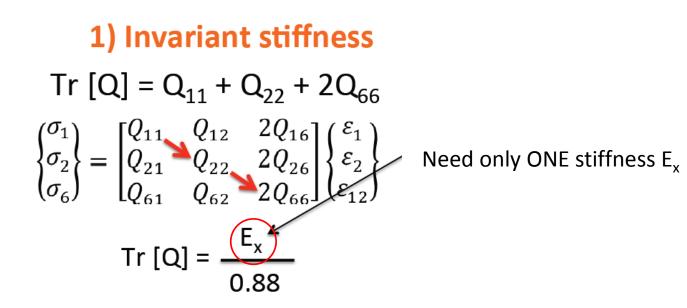


$$Trace = Q_{11} + Q_{22} + 2Q_{66}$$

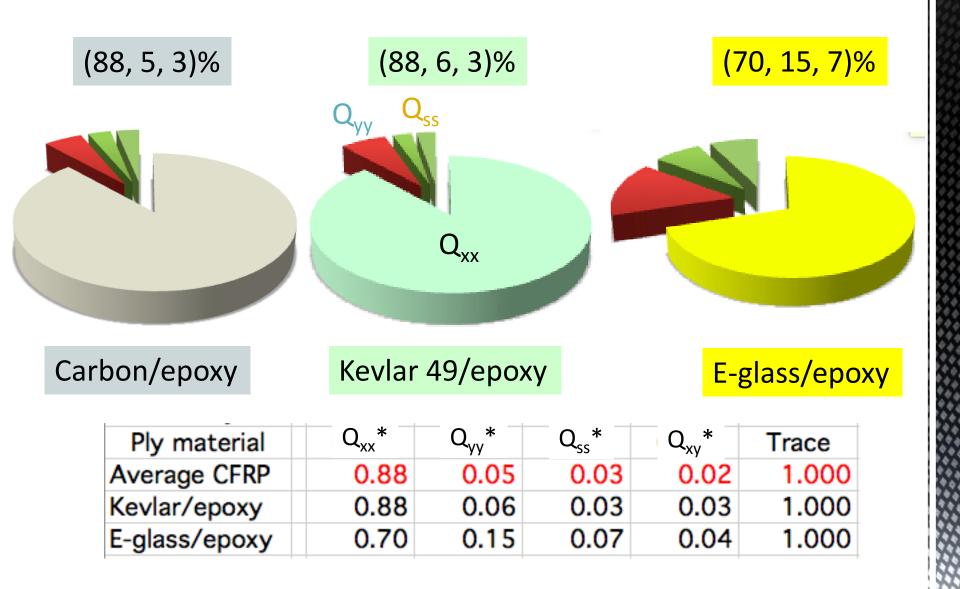
Trace does not vary with ply orientation.

Master Ply

Material	Ex	Ey	νx	Es	Q _{xx} *	Q _{yy} *	Q _{xy} *	Q _{ss} *	Tr	E _x *
			(GPa)				(GPa)			
IM6/epoxy	203	11.20	0.32	8.40	0.8791	0.0485	0.0155	0.0362	232	0.874
IM7/977-3	191	9.94	0.35	7.79	0.8825	0.0459	0.0161	0.0358	218	0.877
T300/5208	181	10.30	0.28	7.17	0.8805	0.0501	0.0140	0.0347	206	0.877
IM7/MTM45	175	8.20	0.33	5.50	0.9014	0.0422	0.0139	0.0282	195	0.897
T800/Cytec	162	9.00	0.40	5.00	0.8955	0.0497	0.0199	0.0274	183	0.888
IM7/8552	159	8.96	0.32	5.50	0.8888	0.0501	0.0160	0.0306	180	0.884
T800S/3900	151	8.20	0.33	4.00	0.9034	0.0491	0.0162	0.0238	168	0.898
T300/F934	148	9.65	0.30	4.55	0.8878	0.0579	0.0174	0.0271	168	0.883
T700 C-Ply 64	141	9.30	0.30	5.80	0.8713	0.0575	0.0172	0.0356	163	0.866
AS4/H3501	138	8.96	0.30	7.10	0.8567	0.0556	0.0167	0.0438	162	0.852
T650/epoxy	139	9.40	0.32	5.50	0.8724	0.0590	0.0189	0.0343	160	0.866
T4708/MR60H	142	7.72	0.34	3.80	0.9029	0.0491	0.0167	0.0240	158	0.897
T700/2510	126	8.40	0.31	4.20	0.8827	0.0588	0.0182	0.0292	144	0.877
AS4/MTM45	127	7.93	0.30	3.60	0.8938	0.0558	0.0167	0.0252	143	0.889
T700 C-Ply 55	121	8.00	0.30	4.70	0.8746	0.0578	0.0173	0.0338	139	0.869
Std dev	24.6	1.0	0.029	1.5	0.0132	0.0053	0.0016	0.0056		0.013
Coeff var %	16.0	10.9	9.0	27.2	1.5	10.1	9.6	17.9		1.5
Master ply					0.8849	0.0525	0.0167	0.0313		0.880



Composition of Trace: Carbon, Kev, Glas



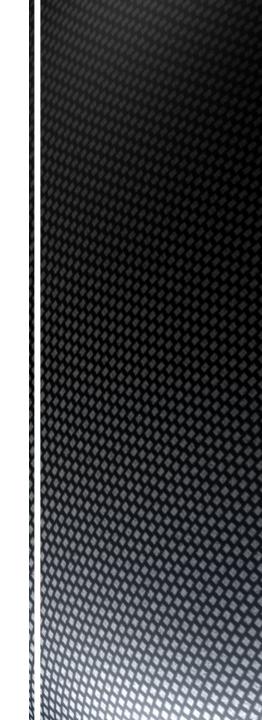
ModuleSelector		X
– MicMac -		1
 In-Pia 	ne: Basic Mic-Mac for in-plane loads	
C In-Pla	ne Strain: Basic Mic-Mac for in-plane straining	
O Duple	x: Mic-Mac for side-by-side comparison of two in-plane loads	
C Thin-v	vall: Sandwich construction with two different faces	
C Fiex: 0	General Mic-Mac for in-plane loads and bending moments	
🔘 GenLa	m: Non-symmetric laminate/sandwich stress analysis	
O PD: Pr	ogressive damage and failure envelopes	
C PDF1e	x. Progressive damage for thin construction and failure envelopes	
C Hybrid	d: Mic-Mac for in-plane loads with two materials	
O Shaft:	Cantilever shaft with torque, distributred and point loads	
O Beam:	Beam with distributred and point loads	
O Tube:	Rectangular and elliptical tubing with distributred and point loads	
C Vesse	1: Thin-walled cylindrical vessel with axial, pressure and torque	
C Lam 3	D	
🗖 Modif	'y Materials? Cancel OK	

MicMac Spreadsheet tool

Calculate macroscopic material properties for any laminate and orientation from microscopic material properties.

How do we get the micro properties

Testing – As Manufactured Material Manufacturer's Data



Mechanical Testing of Composites

Composites Design Workshop X

July 16, 2015

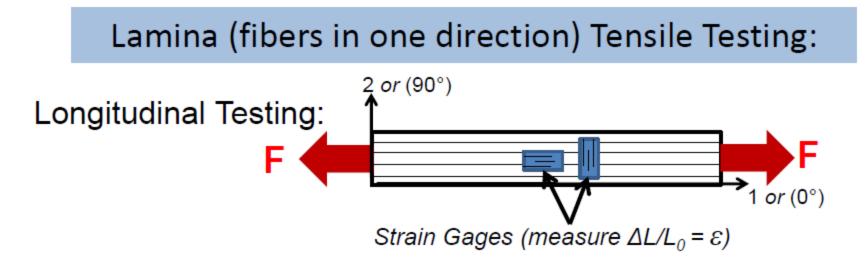
Jared W. Nelson <u>nelsonj@newpaltz.edu</u> SUNY New Paltz, New Paltz, NY

Alan T. Nettles

Stanford University, Stanford, CA NASA Marshall Space Flight Center, MSFC, AL



Tensile Test



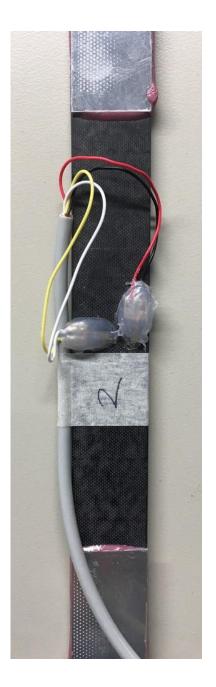
A=cross-sectional area of specimen = width X thickness

Properties typically generated:

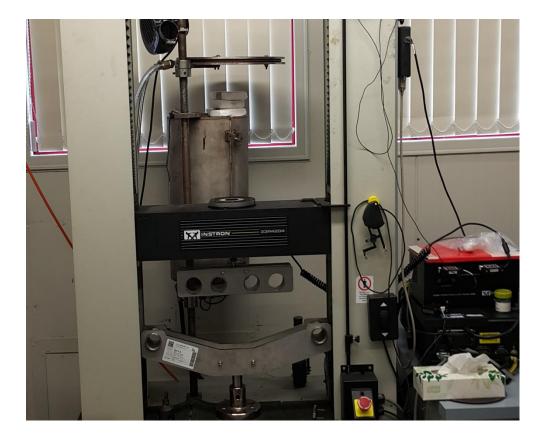
0° Tensile Strength $\sigma_1^{tu} = F^u/A$ Difficult to measure (More on this later)

Longitudinal Tensile Modulus $E_1^t = \sigma_1 / \epsilon_1$

Poisson's Ratio $v_{12} = -\epsilon_2/\epsilon_1 \sim 0.3$



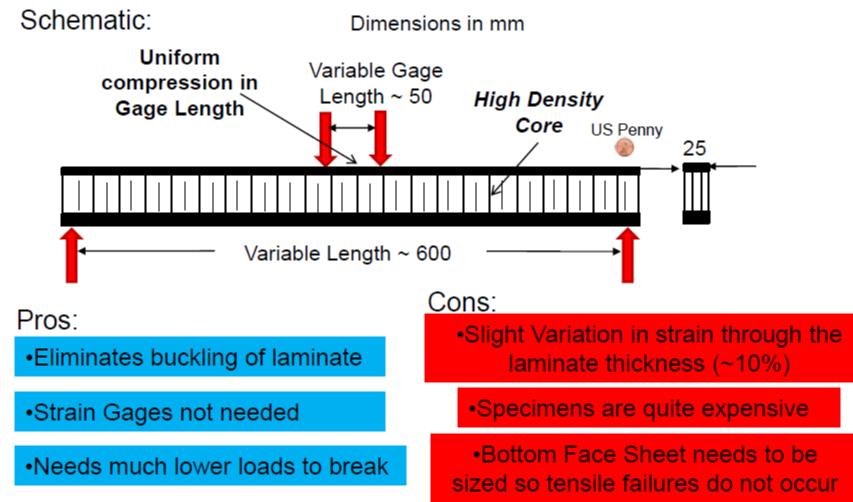
Unidirectional coupon with 0/90 strain gauges and aluminium grips



Four Point Beam Test

Compression Testing Specifics:

ASTM D-5467: Sandwich Beam Method





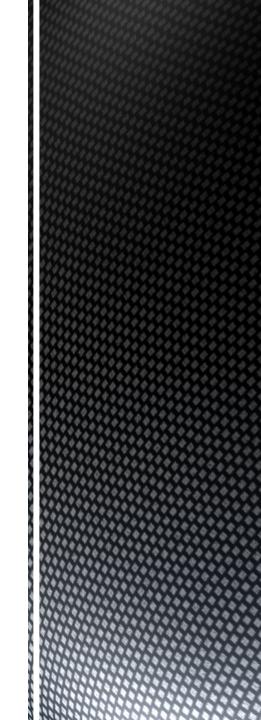
Test Beam

0/90 Carbon Facings Aluminium Honeycomb Core Strain gauges 0/90



Design Allowables

How do we calculate design allowables from test data



Composites Design Workshop-10, July 17, 2015, Stanford University, 1:30 – 2:40 pm (Prof. S. Tsai, Chair)

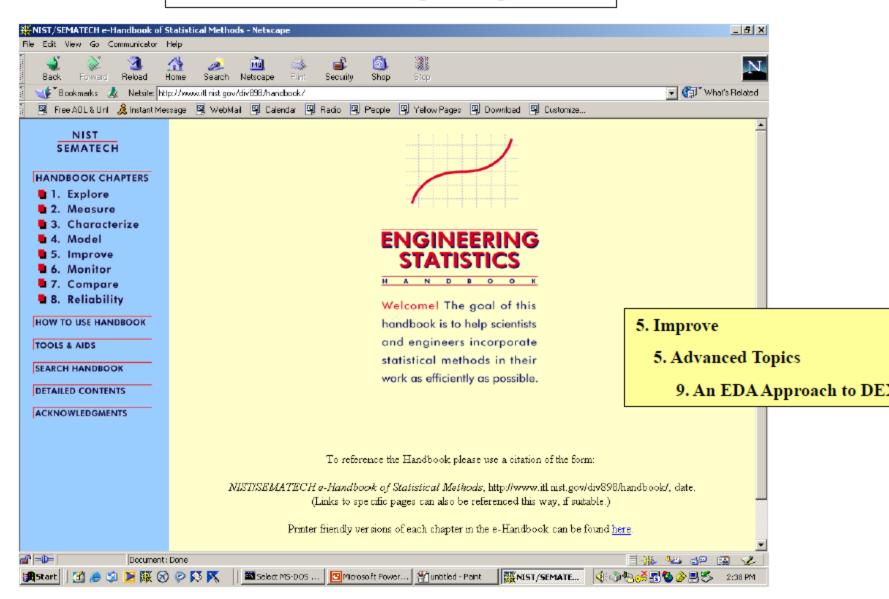


Jeffrey T. Fong, P.E., Ph.D.

Applied & Computational Mathematics Division NIST, Gaithersburg, MD 20899-8910 fong@nist.gov

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NIST e-Handbook of Engineering Statistics



http://www.itl.nist.gov/div898/handbook/ ⁽³⁰⁰

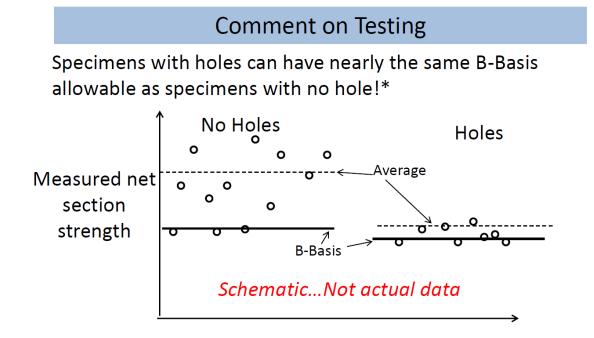
(3000 pages; 3 million page views / month)

Mechanical Testing of Composites Composites Design Workshop X July 16, 2015

Jared W. Nelson <u>nelsonj@newpaltz.edu</u> SUNY New Paltz, New Paltz, NY

Alan T. Nettles Stanford University, Stanford, CA NASA Marshall Space Flight Center, MSFC, AL

A and B basis design allowables depends on test data statistics (JW Nelson)



* First noted to author by by A. Hodge

- As manufactured test samples are a good measure of manufacturing process capability – measure against material manufacturer's data.
- Compression strength of carbon laminates depends highly on configuration – failure mode is buckling of the fibre and depends on lateral support for the fibre.
- Nomex honeycomb core can only support relatively thin carbon fibre face sheets effectively, Al honeycomb is better and can be used in testing.

Our experience with testing

Online, Live Composites Design Workshop XII

June 20-25, 2016; noon to 4 PM PDT; 20 hours + homework US\$1,200 including hardcover and e-books, and tools All sessions recorded/downloadable for individual viewing Widely recognized as the best online training; no travel Optional official transcript of 3 credit CE hours for extra fee Must-learn trace that has revolutionized design and testing



For info/registration: http://compositesdesign.stanford.edu